

Importance of excess oxygen on growth and carrier transport of amorphous In-Ga-Zn-O with impurity hydrogen

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Amorphous oxide semiconductors represented by amorphous In-Ga-Zn-O (a-IGZO) are promising materials for thin-film transistors (TFTs) due to their large field-effect mobilities ($>10 \text{ cm}^2(\text{V}\cdot\text{s})^{-1}$) and small subthreshold voltage swings ($<0.2 \text{ V}\cdot\text{decade}^{-1}$).¹ It has been reported a-IGZO films contain impurity hydrogens and its concentration is varied by the base pressure (P_{base}) of a deposition chamber, which leads to different film structures, densities, and electrical properties^{2,3} and has clarified some hydrogen species work to passivate defects in a-IGZO.⁴ Besides, the roles of hydrogen on subgap states are not understood well. In this work, we fabricated a-IGZO films using two sputtering systems with different P_{base} (denoted STD and UHV sputtering for $P_{\text{base}} \sim 10^{-4} \text{ Pa}$ and $\sim 10^{-7} \text{ Pa}$, respectively) and extensively investigated the hydrogen effects by optical absorption coefficient (α), hard X-ray photoemission spectroscopy (HAXPES), Hall effect measurement, etc.

Figure 1 shows the α spectra of STD and UHV a-IGZO films deposited with various oxygen pressures (P_{O_2}). It is seen that the STD film deposited without O_2 ($P_{\text{O}_2} = 0 \%$) has a long tail subgap absorption extending from the optical band gap ($E_g \sim 3.0 \text{ eV}$) to 0.5 eV , while the UHV films do not show such high-density defect states even though deposited without O_2 . The STD film deposited at $P_{\text{O}_2} = 3 \%$, which is the optimal condition for TFTs, shows the minimum subgap absorption; while, a similar α spectrum with the small subgap absorption is obtained at $P_{\text{O}_2} = 1 \%$ for the UHV a-IGZO. This result quantitatively supports the idea that $\sim 2 \%$ of excess oxygen is required to compensate subgap defects and donors induced by the impurity hydrogens. Figure 2 shows the conductivity as a function of P_{O_2} for STD and UHV films, further supporting this conclusion. Further, HAXPES spectra clarified that the long tail subgap absorption in the $P_{\text{O}_2} = 0 \%$ STD film comes from segregation of metallic In. More details including growth dynamics will be presented at the conference.

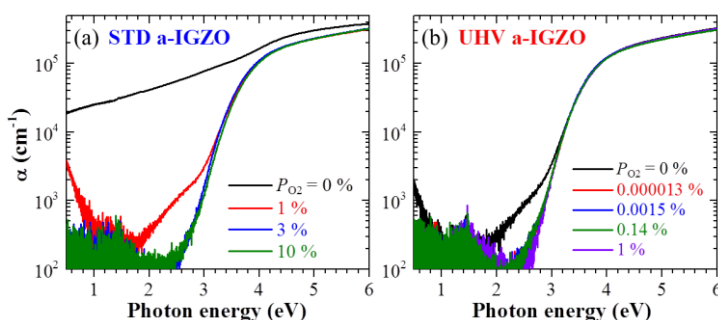


Fig. 1. Absorption spectra of (a) STD and (b) UHV a-IGZO films deposited at various P_{O_2} .

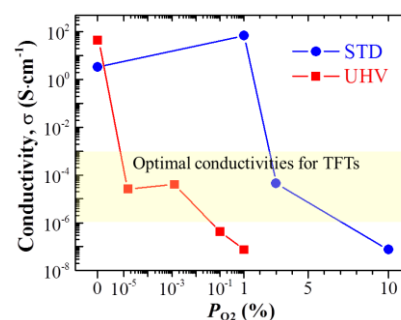


Fig. 2. Conductivity as a function of P_{O_2} for the STD and UHV films.

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